# Hot carrier induced degradation due to multi-phonon mechanism analyzed by lattice and device Monte Carlo coupled simulation

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Abstract – A new multi-phonon model for hydrogen desorption at Si/SiO<sub>2</sub> interface due to hot carriers is proposed for a multi-scale simulation, in which Lattice Monte Calro method is coupled with Device Monte Calro method by using a mediator-based common software platform. The power law between interface trap density and time  $(N_{tt} \propto t^{\alpha})$  is demonstrated which shows good agreement with experimental results. It is shown that the multi-phonon mechanism has significant effect on reliability of MOS devices with ultra-short gate under low applied voltage.

#### I. INTRODUCTION

Hot carrier damage on gate dielectrics resulting in V<sub>th</sub> shift and current degradation is a barrier to achieving high performance in ultra-small MOS devices[1]. Experimental results of the performance degradation show an isotope effect[2],[3]. This indicates that mechanisms of defect generation are associated with phonon energy of hydrogen atoms at Si/SiO2 interface[4],[5]. To study the defect generation due to hot carriers, a device simulation used to analyze non-equilibrium carrier transport under a 100nm gate electrode needs to be incorporated with atomic-scale phenomena on gate dielectrics. Time scale of the defect generation varies from seconds to million-seconds. In contrast, time required for carriers propagating through the gate electrode requires less than pico-seconds. Moreover, the defects at the interface will affect the carrier distribution that determines number of electrons scattered by H phonon, which in turn determines desorption rate of H atoms at the interface. In this paper, a new H desorption model due to multi-phonon mechanism is proposed to study the stochastic desorption rate of H atoms determined by hot electron distribution. Lattice Monte Carlo simulation (LMC) is coupled with device Monte Carlo simulation (DMC) on the Hitachi parallel machines and workstation clusters. A mediator-based common software platform, described in an earlier paper[6], is employed to efficiently combine different kinds of simulation methods to a single simulation. The coupled simulation can be utilized to analyze the intertwined phenomena with varying scales in space and time.

# II. MULTI-SCALE SIMULATION FLOW

Fig. 1 shows an analytical region near Si/SiO<sub>2</sub> interface under a gate electrode, in which LMC is coupled with window-DMC hybridized with continuum simulation of the whole device region. For instance, a device with gate length  $L_{\rm g}\!=\!240{\rm nm}$  and oxide thickness  $T_{\rm ox}\!=\!5{\rm nm}$  is used for DMC of which analytical region is  $190{\rm nm}\times10{\rm \mu m}\times250{\rm nm}$  under effective gate length in the Si substrate.  $5\times10^4$  particles are used in DMC. Boundary conditions of potential profile and carrier distribution for DMC are obtained by the continuum simulation. Analytical region for LMC is the surrounding region near Si/SiO<sub>2</sub> interface of  $250{\rm nm}\times300{\rm nm}\times3{\rm nm}$  in which periodic boundary condition is applied to width direction. The analytical region is discretized by uniform meshes of  $10{\rm \AA}$ . H atoms and defects exist at the interface according to initial density.

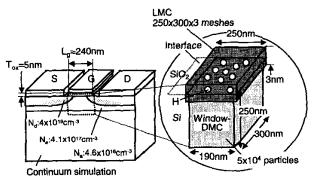


Fig. 1 Analytical region associated with L<sub>s</sub>, in which lattice Monte Carlo (LMC) is coupled with device Monte Carlo (DMC) simulation

A flow chart of spatial and temporal multi-scale simulation is shown in Fig. 2. LMC provides H atoms and defect configuration ft to DMC using the mediator-based platform according to correlative specification between particles and meshes. The in-sphere correlation is employed to transform the charge of defects in LMC to charge distribution associated with correlated mesh points inside of the sphere centered at the defects in DMC. The physical values are transformed and transferred automatically by mediators associated with each simulation. In DMC, real scattering of H atoms and defects as well as the conventional phonon and impurity scattering are taken into account. The time step  $\Delta t^D$  in DMC is set to 1fs. Number of electrons scattered by H atoms at the interface and their respective

energy distributions f<sup>D</sup> are accumulated in DMC for 5 ps and sent to LMC. The correlative specification of the nearest point is used to transform the number of electrons and the energy distribution to a lattice point which is the nearest point from the electrons. These physical values are used to analyze the desorption rate of H atoms in LMC based on the new multi-phonon model. The temporal distribution of H atoms and defects at the interface can be obtained by solving rate equations for the defect generation using Monte Carlo method. The time step  $\Delta t^L$  of LMC is set to more than 1ms. After forming new configuration of H atoms and defects, the physical values are transformed and transferred to DMC. The procedure is executed iteratively to communicate new configurations of H atoms, defects and electron distribution, in which LMC and DMC are executed simultaneously on the parallel machines and workstation clusters.

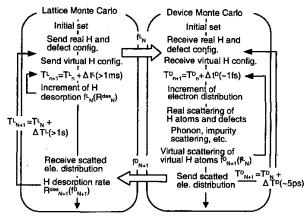


Fig. 2 Flow chart of multi-scale simulation

### III. ACCELERATION ALGORITHMS

Since the coupled simulation has different spatial and time scales, new algorithms are proposed to accelerate execution time of the multi-scale simulation. The following acceleration algorithms have been implemented.

## 1) Virtual scattering of electrons at H atoms in DMC

Besides the real scattering due to H atoms which exist at the interface randomly, virtual H atoms with density  $10^{13} \rm cm^2$  are configured uniformly. Electrons inside of a virtual scattering radius  $r_s^v$  centered at virtual H atoms with high scattering rate coefficient  $R_s^v$  are assembled and allowed to undergo free flight. These virtually scattered electrons are sent to LMC and scaled to calculate the desorption rate according to a real scattering radius  $r_s^v$  and a real scattering rate  $R_s^v$ . For instance,  $r_s^v$ =2.5Å,  $R_s^v$ =3x10<sup>11</sup>s<sup>-1</sup> and  $r_s^v$ =10Å,  $R_s^v$ =10<sup>14</sup>s<sup>-1</sup> are used. The execution time is accelerated by 10<sup>4</sup> times faster, as the number of scattered electrons is proportional to be  $r_s^{3*}R_s^v$ 

## 2) Replication of an analytical region in LMC over DMC

The periodic boundary condition is applied to the analytical region in LMC over the entire Si/SiO<sub>2</sub> interface in width direction. This reduces total number of real H atoms to solve rate equations for the defect generation in LMC. Furthermore, using replication of the analytical region, the execution time to assemble the virtual scattered electrons in the whole region in DMC can be reduced according to total number of the virtual H atoms.

#### 3) Varying time step for H desorption in LMC

To solve rate equations for H desorption in LMC, the time step can vary according to desorption rate of H atoms. Based on the H desorption rate of multi-phonon mechanism, time required for 5% desorption of total number of H atoms for each iteration cycle  $\Delta T^L$  is evaluated and divided equally into  $10^3$  time steps of  $\Delta t^L$ . Time required for H desorption varies from seconds to more than million-seconds. Therefore, time step in LMC is set to few milliseconds in the first few cycles and increase to few thousands of seconds in later cycles according to the desorption rate. This technique is very significant in view of appropriate load balancing in multiscale coupled simulations.

#### IV. Multi-Phonon Model

Under given bias condition, electrons propagating through the gate can be scattered by H atoms at the interface and the H atoms will absorb and emit the phonon leading to excited and de-excited states. When the excited phonon state reaches to the threshold energy, the H atoms undergo desorption and dangling bond gets created which becomes interface traps.

Excitation and de-excitation of the phonon states of H atoms depend upon phonon energy of H atoms and kinetic energy of scattering electrons. To mask complexities involved in each scattering event, the typical H-Si bending phonon energy is taken into account. Total number of the scattered electrons is separated into two categories. Emission scattering electrons, N<sub>w</sub> are number of electrons capable of emitting phonon after scattering with H atoms and average energy of emission scattering electrons is denoted by E<sub>u</sub>. Similarly, absorption scattering electrons, N<sub>d</sub>, are number of electrons capable of absorbing energy after scattering with H atoms and average energy of absorption scattering electrons is denoted by E<sub>4</sub>. The electrons possessing kinetic energy greater than the typical phonon energy are capable to excite H atoms. Whereas, all scattering electrons are capable to de-excite H atoms, provided H atoms are not in the ground state. Fig.3 and Fig.4 show dependence of N<sub>u</sub>, N<sub>d</sub> and E<sub>u</sub>, E<sub>d</sub> on the position under the gate electrode, respectively. N<sub>d</sub> has a sharp peak structure near source side. Number of electrons at the source side is high and all scattering electrons can de-excite the phonon state. N<sub>u</sub> has a small peak at the middle region of the gate. The electron energy near source side is too low to excite H atoms. As electrons are accelerated along the gate, both E. and E<sub>d</sub> have peak near the drain side. The lowest energy of E<sub>m</sub> that can excite H atoms at the source side is almost same as typical phonon energy.

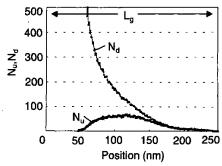


Fig.3 Dependence of N<sub>u</sub>, N<sub>d</sub> on position

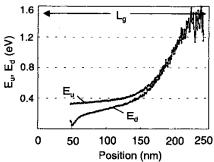


Fig.4 Dependence of Eu, Ed on position

Effective probability of excitation from lower to higher phonon level, Pu(Nu, Eu), and probability of de-excitation from higher to lower level, P<sub>d</sub> (N<sub>d</sub>, E<sub>d</sub>), are obtained by averaging probability of excitation and de-excitation over all phonon level of H atoms and number of scattering events m, respectively.

$$P_{u}(N_{u}, E_{u}) = \frac{\Sigma_{n} f(m,n) \pi_{u,n}(N_{u}, E_{u})}{\Sigma_{n} f(m,n)}$$
(1)
$$P_{d}(N_{d}, E_{d}) = \frac{\Sigma_{n} f(m,n) \pi_{d,n}(N_{d}, E_{d})}{\Sigma_{n} f(m,n)}$$
(2)

$$P_{d}(N_{d}, E_{d}) = \frac{\sum_{n} f(m, n) \pi_{d,n}(N_{d}, E_{d})}{\sum_{n} f(m, n)}$$
(2)

Where, f (m, n) is the number of pathways from 0 to n phonon levels at  $m_{th}$  scattering event and  $\pi_{un}$   $\pi_{dn}$  are normalized probabilities for excitation and de-excitation at n<sub>th</sub> phonon excitation level, respectively. While calculating  $\pi_{d,n}$ , relaxation time for H phonon,  $\tau_{relax}$ , is taken into account. Typical relaxation time for  $\tau_{relax}$  is 10 ns[5]. The desorption probability of H atoms due to the multi-phonon mechanism can be derived by the following expression.

$$P_{des} = \sum_{m} f(m,n) P_{u}^{(m+n)/2} (E_{u}, N_{u}) P_{d}^{(m-n)/2} (E_{d}, N_{d})$$
 (3)

The threshold energy, Eth=3.7eV, for H bond breaking and the H-Si bending phonon energy, ω<sub>0</sub>=0.29eV, are used. Hence, the number of phonon excited level, n, becomes 13 for H desorption and 18 for deuterium desorption resulting in the isotope effect. As shown in Fig. 5, partial desorption probability shows finite stochastic probability at scattering events at m = 13, 15, 17, ----. For instance, m=17 consists of 15 times of excitation and 2 times of de-excitation resulting in H desorption. Total desorption probability can be obtained by summing up all partial desorption probability and saturates to a constant value  $P_{\text{des}}$  after time  $\tau_{\text{des}}$ . As a consequence, the desorption rate of H atoms, R<sub>des</sub>, can be obtained by dividing  $P_{des}$  by  $\tau_{des}$ .

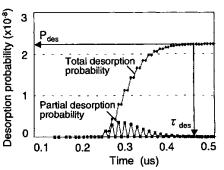


Fig. 5 Total and partial desorption probability at position of 150nm

# V. RESULTS AND DISCUSSION

In Fig. 6,  $R_{des}$  as well as  $P_{des}$  and  $\tau_{des}$  are shown to be strongly dependent on the position under the gate electrode.

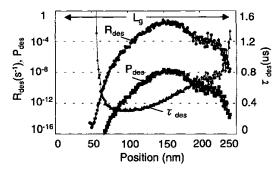


Fig. 6 Desorption rate Rdes and probability  $P_{des}$  and time  $\tau_{des}$ 

Maximum R<sub>des</sub> appears at the middle region. This is because N<sub>u</sub> becomes large and E<sub>u</sub> is higher than E<sub>d</sub> at the middle of the gate leading to high Pdes. Whereas, the Rdes decreases near source side, as N<sub>d</sub> becomes high leading to low  $P_{des}$  and  $\tau_{des}$  becomes long in the source region. In the drain side, R<sub>des</sub> and P<sub>des</sub> have a slight flat structure and decrease

steeply, where the phonon relaxation time becomes shorter than the typical time for excitation.

Based on the multi-phonon desorption rate, the coupled simulation is applied to the device with  $L_g=240$ nm and  $T_{\rm ox}=5$ nm. In Fig. 7, the universal power law  $N_{\rm ii} \propto t^{\rm a}$  is shown to be in good agreement with experimental results of  $L_g=350$ nm and  $T_{\rm ox}=5.5$ nm, where the stress bias of  $(V_g-V_{\rm th})/T_{\rm ox}$  is same in order to set both of the electron concentration to be equal. Power of defect generation is obtained by straight line in fitting the region 2 s to  $10^3$  s, of which power  $\alpha$  is 0.51. Please note that the value of power is a direct consequence of positional dependence of the desorption rate shown in Fig.6. Initial portion of the power law is associated with H desorption at the middle region of the gate, whereas saturation region observed after  $10^3$  s is associated with H desorption near the source and drain region.

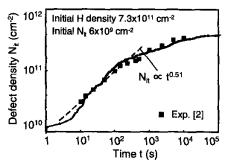


Fig. 7 Power law between defect density and time at stress condition of  $(V_q^- V_{th})/T_{ox} = 0.15(V/nm), V_q = 3V$ 

The coupled simulation is also applied to an ultra-small device with  $L_g$ =13nm,  $T_{ox}$ =1nm and halo p-doping profile obtained from MIT web-site[7]. The power law of 13nm gate device is compared to that of 240nm in Fig.8.

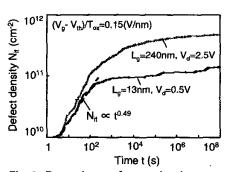


Fig. 8 Dependence of power law between defect density and time on  $\mathbf{L}_{\mathbf{g}}$ 

 $L_g$ =13nm and  $L_g$ =240nm is set to be 0.5V and 2.5V predicted in ITRS'01, respectively. It is found that the interface traps are generated under low  $V_d$  of 0.5V in  $L_g$ =13nm device, of which power  $\alpha$  is 0.49 for a short period up to  $10^2$  s. The power decreases after long current stress because  $N_u$  becomes too small to lead to H desorption at the source side, of which electron energy  $E_u$  needs to be higher than the typical H phonon energy. This shows that hot carrier damage due to the multi-phonon mechanism will occur slowly in the future device when low operation voltage is higher than H phonon energy.

#### VI. CONCLUSION

Multi-scale simulation, in which Lattice Monte Carlo is coupled with device Monte Carlo method, has been employed to analyze defect generation due to hot electrons. The new multi-phonon model and new acceleration algorithms have been proposed to realize space and time scalability. The power law between interface traps and time obtained from the coupled simulation shows good quantitative agreement with experimental results. It is found that 10nm gate device with low operation voltage still has damage on dielectrics but occurs slowly. The mediator-based coupled simulation allows cooperative designs for high-performance and high-reliability. It also allows to efficiently enhancing complex interaction mechanisms such as H diffusion and tunneling current by communicating H atoms, defects and electron distribution in dielectrics.

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 $(V_g-V_{th})/T_{ox}$  is set to be same. The operation bias of  $V_d$  for