Transport-reaction model for interface state build-up at the Si-SiO₂ interface

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Abstract

The MOS interface state build-up process was simulated for a 1D MOS structure within a framework of reaction-transport model. The identified kinetics is summarized as simple power laws. The transport-limited mode of half power of the stress time has emerged universally with regardless of the charge state of transported species.

1 Introduction

Oxide reliability has been an issue since the advent of MOSFETs. Several reaction-transport models have been reported dealing with a 1D MOS problem (Fig. 1) [3, 5, 4]. Their results look somewhat contradictory, as highlighted in Figs 3 and 4. Particularly, the nature of the transport-limited mode is inconclusive. In experiment, on the other hand, Doyle *et al.* identified a universal 0.5 power law of the stressing time or the amount of injected charge[2]. This situation has motivated us to streamline theoretical understanding. Exploring comparison with measurement is beyond the scope of this work.

2 Simulated Problem

Interface state(IFS) build-up was simulated within a framework of the surface reaction (interface state generation and passivation) and transport phenomena of hydrogen-related species. The rate equation for the surface reaction and the transport equation for the hydrogen-related species were numerically solved with an aid of a partial differential equation solver PROMIS[1]. For simplicity, we have stood back on the simplest situation, a MOS structure as a one dimensional(1D) problem, as illustrated in Figs. 1 and 2.

3 Results

3.1 IFS kinetics—neutral species involved

Simulation has revealed that the interface state(IFS) build-up process obeys a set of power laws of the stressing time, as shown in Fig. 5. In the course of stressing in Fig. 5, the identified power laws are t^1 (reaction limited), t^0 (quasi-equilibrium), $t^{1/4}$ (diffusion limited

mode which fits in Jeppson's solution), and $t^{1/2}$ (another diffusion-limited mode). They fall into two groups; a transport-sensitive mode and a transport-*in*sensitive one. The latter two modes of $t^{1/4}$ and $t^{1/2}$ are D_H (diffusivity of the diffusing species in the oxide)-sensitive, as shown in Fig. 6. The $t^{1/2}$ mode is also T_{ox} (oxide thickness)-sensitive. The $t^{1/4}$ mode does not reveal a T_{ox} dependence, as shown in Fig. 7. As for neutral diffusing species, power laws are readily summarized in Table 1, most of which can be confirmed also through analytical solution owing to the simplicity of the 1D problem.

3.2 IFS kinetics —charged species involved

When charged species are transported, the presence of oxide electric field(E_{ox}) creates an asymmetricity in IFS kinetics, as shown in Figs. 8–10. When E_{ox} is repulsive ("+Vg" case in Fig. 8), IFS build-up slows down. When E_{ox} is attractive ("-Vg" case in Fig. 9), the process is accelerated, with its $t^{1/2}$ relation unchanged. The $t^{1/2}$ mode is E_{ox} -sensitive and branches off at an earlier period as E_{ox} becomes strong. In the $t^{1/2}$ mode, IFS concentration was found to be proportional to $E_{ox}^{1/2}$ (Fig. 10), in contrast to Ogawa[5].

The $E_{ox}^{1/2}$ dependence was derived also from analytical solution, as summarized in Fig. 11, which supports our numerical simulation and disproves Ogawa's asymptoic solution. Note that the E_{ox} , T_{ox} , and D_H dependence on IFS is built in the transport coefficient $A_I^{1/2}$. The $E_{ox}^{1/2}$ dependence occurs in the strong E_{ox} limit. In this limit, an explicit T_{ox} dependence vanishes. In the opposite limit of weak E_{ox} , our analytical solution reduces to that for neutral species, which agrees with Hu's solution[3].

3.3 Lifetime—sublinearity leverage

Lifetime τ , the time which it takes for IFS to reach a threshold, is leveraged by the sublinear power of time. In the transport-limited mode, the threshold IFS concentration is proportional to $E_{ox}^{1/2}\tau^{1/2}$. Solving back this relation with respect to τ , the lifetime becomes $\tau \propto E_{ox}^{-1}$. When E_{ox} decreases by a factor of 1/10, IFS build-up slows down by a factor of $\sqrt{10}$, but the lifetime increases by a factor of 10. Seeking a less stringent stressing condition, hence, deserves its efforts.

4 Conclusion

We applied the reaction-transport model to a 1D MOS structure. The obtained IFS kinetics can be summarized as simple power laws. The transport-limit mode of $t^{1/2}$ was elaborated using numerical and analytical solutions. Irrespective of neutral or charged species, the transport-limited mode of $t^{1/2}$ universally occurs. It shows the $E_{ox}^{1/2}$ dependence for charged species.

References

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Table 1: Power law for IFS build-up with neutral diffusing species involved. Exponents for various quantities are listed for "Reaction limited", "Equilibrium", and "Diffusion limited(two types)" modes. In the "2nd diffusion-limited mode," for example, N_{IFS} is proportional to $t^{1/2}, T_{ox}^{-1/2}, (k_G/k_R)^{1/2}, D_H^{1/2}, C_{SH,0}^{1/2}$. †denotes N_{IFS} is proportional only to k_G rather than the ratio.

Mode	Time t	Oxide thick-	Rate constant	Diffusivity	Initial Si-H
		ness T_{ox}	ratio (k_G/k_R)	D_H	conc. C_{SH}
Reaction limited	1	0	†	0	1
Eqlib.	0	0	1/2	0	1/2
1st diff. limited	+1/4	0	1/2	+1/4	1/2
2nd diff. limited	+1/2	-1/2	1/2	+1/2	1/2

SizH Gate oxide	$\frac{\partial C_{IFS}}{\partial t} = k_G C_{SH} - k_R C_{IFS} C_H$	(surface reaction)(1)		
Si - H- Gate	$\frac{\partial C_H}{\partial t} + \operatorname{div} \vec{J}_H = G - R$ $J_H = -D_H \nabla C_H + \mu_H C_H E_{ox}$	(hydrogen transport) (hydrogen flux)	(2) (3)	
Si–H ⇔ Si* + H Interface state generation / passivation Transport of hydrogen–related species	$G - R = k_G C_{SH} - k_R C_{IFS} C_H$ G - R = 0	(surface) (bulk)	$(4) \\ (5)$	



C _H : concentration of hydrogen species C _{IFS} : interface state concentration		•		
N _{IFS} : interface state areal density, N _{IFS} ~L _R C _{IFS} Assumption: L _R ~ 1nm(reaction region), C _{IFS} (X)~constant for X within L _R C _{SH} : SiH (IFS precursor) concentration, C _{SH0} : initial C _{SH} J _H : hydrogen flux areal density D _H : diffusivity for hydrogen species in oxide $\mu_{\rm H}$: mobility of hydrogen species in oxide connected with D _b . by Einstein relation	$\begin{tabular}{ c c c c } \hline C. \ Hu[2] \\ \hline N_{IFS}(t)^{\infty} t^{1/2} \\ diffusion limit \\ finite \ T_{ox} \\ (sink \ b.c.) \end{tabular}$	Jeppson[3] N _{IFS} (t)∝ t ^{1/4} diffusion limit infinite T _{ox}	$\begin{array}{c} Ogawa[4] \\ \hline N_{IFS}(t) \simeq t^{1/4} t^{1/2} \\ transport limit \\ inifinite T_{ox} \\ finite T_{ox} \\ (sink \ b.c.) \end{array}$	$\label{eq:transformation} \begin{array}{c} \mbox{This work} \\ \mbox{N}_{\rm IFS}(t) \propto t^{1/4}, t^{1/2} \\ \mbox{transport limit} \\ \mbox{finite } T_{\rm ox} \\ \mbox{(sink b.c.)} \end{array}$
T _{ox} : oxide thickness E _{ox} : electric field across the oxide (oxide field) k _G :k _R : surface reaction rate constants for IFS generation and passivation (constant in this work) k _G includes hot carrier flux.	neutral analytical t: stressing tim	neutral analytical ne; sink b.c. = sink t	neutral ion analytical (asymptoic) poundary condition at	neutral ion numerical ox/gate boundary

Figure 2: Notations.

Figure 3: Previous works and this work.



positively charged species; repulsive e-field.

positively charged species; attractive e-field.

Figure 10: IFS kinetics with positively charged species; IFS concentration at $t=10^8$ unit.



